known to react with organometallic reagents, and the synthetic chemical literature abounds with examples of removable protecting groups.

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# **Surface Reaction of Poly[ bis( trifluoroethoxy) phosphazene] Films by Basic Hydrolysis**

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**<sup>A</sup>**novel surface reaction of a fluoroalkoxyphosphazene polymer is reported. Films of poly[bis(tri- fluoroethoxy)phosphazene], [NP(OCH,CF,),],, underwent surface hydrolysis when treated for **0-15** min with 8 M aqueous sodium hydroxide in the presence of **0.05** M tetra-n-butylammonium bromide **as** a phase-transfer agent. This reaction converted a hydrophobic surface to one with hydrophilic and adhesive character. No surface reaction by NaOH solutions was detected in the absence of  $Bu_4NBr$ . On the basis of a series of surface analytical data, the principle chemical reaction was found to be a replacement of P-OCH<sub>2</sub>CF<sub>3</sub> side groups at the polymer surface by P-O<sup>-</sup>NBu<sub>4</sub><sup>+</sup> units. The modified surfaces were found to possess anionic sites, since an exchange reaction with the cationic dye methylene blue took place readily at the surface. No exchange occurred with the anionic dye Biebrich scarlet. Microscopic examination of the dyed films allowed the depth of the surface reaction to be measured. Contact-angle analysis showed that the surface of the polymer became progressively more hydrophilic as the reaction progressed. No signifcant changes in surface texture were deteded. Surface reactions were **also** *carried* out with a commercial mixed-substituent fluoroalkoxyphosphazene elastomer. After extensive surface hydrolysis, films of [NP-  $(OCH_2CF_3)_2$ , retained their solubility in organic solvents, in which they were analyzed by <sup>31</sup>P, <sup>13</sup>C, <sup>1</sup>H, and <sup>19</sup>F NMR spectroscopy. The NMR analysis also suggested the presence of -O<sup>-</sup>NBu<sub>4</sub><sup>+</sup> units. GPC analysis of an extensively modified material indicated that some cleavage of the phosphazene backbone may have occurred. The hydrolysis process was monitored by the reaction of solutions of  $\mathrm{[NP(OCH_2CF_3)_2]}$ , with NaOH in the presence of  $Bu_4$ NBr in refluxing THF. The mechanism of adhesion is discussed.

#### **Introduction**

**A** need exists to understand more fully the physical and chemical processes that are responsible for polymer adhesion and biocompatibility.<sup>1,2</sup> It is known that fluorinated polymers such as poly(tetrafluoroethy1ene) or poly(chlorotrifluoroethy1ene) are difficult to bond to other materials because of their hydrophobic surfaces. Surface reactions of fluoropolymers to enhance adhesion have been studied extensively. $3.4$  In most cases, it is critical that the surface reaction should not affect the bulk properties of the polymer and that the depth of the reaction zone can be controlled easily.

Poly(organophosphazenes)<sup>5-8</sup> are a relatively new class of polymers that are of increasing scientific and technological interest. **A** typical two-step synthesis of a poly- (organophosphazene) is illustrated in Scheme **I.** It involves a ring-opening polymerization of hexachlorocyclo-

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**(8) Allcock, H. R. Phosphrous-Nitrogen Compounds; Academic: New York, 1972.** 



triphosphazene **(1)** at **250 "C** to form poly(dich1orophosphazene) **(2).** Polymer **2** functions **as** a macromolecular intermediate. The high reactivity of the P-Cl bonds allows the polymer to undergo facile nucleophilic substitution reactions with sodium aryl oxides or alkoxides or with primary or secondary amines. The resultant poly- (organophosphazenes) are usually stable to air and moisture. **A** wide variety of different types of side groups can be incorporated into the polymer structure, with the bulk and surface properties being dependent on the side groups present. The reactions and structures of cyclic phosphazene molecules such as **1** are useful models for the reactions and structures of the analogous macromolecules? Despite their technological utility, the surface chemistry

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<sup>(4)</sup> Dias, A. J.; McCarthy, T. J. Macromolecules 1984, 17, 2529.<br>
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(b) Allcock, H. R.; Kugel, R. L.; Valan, K. J. Inorg. Chem. 1966, 5, 1709.<br>
(c) Allco

**<sup>(9)</sup> Allcock, H. R. Acc. Chem. Res. 1979,12, 351.** 



of **poly(organophosphazenes)** has not yet been investigated in detail.<sup>10</sup>

**Poly[bis(trifluoroethoxy)phosphazene] (3)** is prepared by the reaction of **2** with sodium trifluoroethoxide. Polymer **3** is a semicrystalline thermoplastic that is soluble



in common organic solvents such as tetrahydrofuran (THF) or acetone. The surfaces of films of **3** are hydrophobic and show no adhesive character. Commercial fluoroalkoxyphosphazene elastomers **(4)** possess a chemical structure similar to that of **3** but contain both trifluoroethoxy and longer chain fluoroalkoxy side groups.11 Many current or potential applications of **4** involve its contact with metals, ceramics, polymers, or fabrics. Hence, the surface character of **4** plays an important role in its use in a variety of applications. The surface modification of **3** and **4** to enhance ita adhesion is **also** relevant to potential biomedical applications of phosphazene polymers.

Here we report that the hydrophobic surfaces of **3** and **4** can be modified by treatment with hot aqueous solutions of sodium hydroxide in the presence of tetra-n-butylammonium bromide as a phase-transfer agent to form hydrophilic and adhesive surfaces. This surface reaction is simple and rapid. Furthermore, the rate of the reaction can be controlled by several experimental parameters.

Previous research has shown that the cyclic analogue of **3,** [NP(OCH2CF3)2]3 **(5),** is hydrolyzed in aqueous organic sodium hydroxide solutions (Scheme II).<sup>12</sup> The hydrolysis reaction of **5** involves a nongeminal replacement of the trifluoroethoxy side groups by hydroxyl side groups. The first replacement is more facile than the second. Similar hydrolysis reactions were reported for the cyclic tetrameric analogue,  $[NP(OCH_2CF_3)_2]_4$ . However, films of polymer **3** are stable to hydrolysis when treated with aqueous sodium hydroxide, even at high temperatures and after prolonged reaction times.

It has recently been reported that the physical properties of **3** are slightly dependent on the exact experimental procedure used to prepare the polymer.<sup>13,14</sup> Some evidence exists that these differences in properties are due to the presence of small amounts of hydroxyl side groups generated during the reaction of **2** with sodium trifluoroethoxide or during the purification of the polymer.

The following questions were addressed in this investigation: **(1)** What types of reactions occur at the surface of **3** following its immersion in solutions of NaOH/ Bu4NBr? For example, does a replacement of trifluoroethoxy side groups by hydroxyl groups *occur* at the surface of **3** in a manner similar to the reactions of the model compound **5? (2)** Can the reactions be monitored by surface analytical methods such **as** depth profile, XPS, contact-angle, or ATR-IR analyses? **(3)** Can the modified polymer surfaces undergo surface reconstruction when exposed to solvent environments of different polarity or when heated? **(4)** If the surface reaction is allowed to proceed long enough, will the modified polymer still remain soluble in organic media to allow analysis by solution methods such **as NMR** spectroscopy or gel permeation chromatography (GPC)? **(5)** Does the solution hydrolysis of **3** by NaOH in the presence of Bu4NBr in THF model the surface reactions of **3? (6)** Do mixed-substituent fluoroalkoxyphosphazene elastomers **(4)** and other poly- (organophosphazenes) undergo similar surface reactions when treated with aqueous NaOH/Bu<sub>4</sub>NBr solutions? (7) What is the mechanism of adhesion for these modified polymer surfaces?

## **Results and Discussion**

**Qualitative Aspects of the Surface Reaction of 3 with Base.** Polymer **3** was prepared by a method reported earlier<sup>5</sup> (Scheme I) with the use of a recent modification.<sup>13</sup> Smooth opalescent films of 3, 140  $\mu$ m thick, were prepared on glass plates by solution-casting methods. The films possessed no adhesive character before the surface reaction, and scanning electron microscope **(SEM)** analyses showed that the films were smooth on both the glass and **air** sides. Although **3** is a microcrystalline polymer, **SEM**  analyses did not distinguish the crystalline regions from the amorphous regions at the polymer surface.<sup>15</sup> Gentle scraping of the films did not remove any of the polymeric material from the surface. The films were washed with hexane before use to remove any contaminants (e.g., organosiloxanes from stopcock grease) that may have migrated to the surface.

Films of 3 were immersed in 8 M NaOH/0.05 M Bu<sub>4</sub>NBr solutions at 80  $\degree$ C. After several minutes of immersion, the treated films were removed from the hot base solution and were rinsed with distilled-deionized water for several minutes to remove loosely adsorbed species. Finally, the films were dried under vacuum.

The surface-reacted films of **3** were pressed onto glass, stainless steel, aluminum foil, ceramic, polymer, or human skin substrates and were found to adhere to each of these materials. Attempts were made to peel the films from these substrates with forceps, but both visual and **SEM**  analyses indicated that some of the polymer remained on the substrates (cohesive failure). The peel strength **of**  surface-modified **3** pressed onto aluminum foil by hand at room temperature was 20 N/m. As a comparison, the peel strength of a typical consumer-grade Scotch tape pressed onto aluminum foil by hand at room temperature was 80 N/m, measured under the same conditions as for modified **3.** 

**<sup>(10)</sup> For example, see: Allcock, H. R.; Kwon,** *S. Macromolecules* **1986,**  *19,* **1502.** 

**<sup>(11)</sup> See:** *EYPEL-F, Performance Elastomers, Technical Data;* **Ethyl (12) Allcock, H. R.; Walsh, E. J.** *J. Am. Chem. SOC.* **1972,** *94,* **119. Corp., Baton Rouge, LA, 70801,1987.** 

**<sup>(13)</sup> Ferrar, W. T.; Marshall, A. S.; Whitefield, J.** *Macromolecules*  **1987, 20, 317.** 

**<sup>(14)</sup> Ferrar, W.; et al.** *Phosphorous, Sulfur, Silicon* **1989,** *41,* **147. (15) Blais, P.; Carlsson,** D. **J.; Csullog, G. W.; Wiles,** D. **M.** *J. Cotloid Interface Sci.* **1974,** *47,* **637.** 



The films were still opalescent after the surface reaction, and gentle scraping of the surface revealed the presence of a gellike adhesive material at the surface. When films of **3** were floated on the 8 M NaOH/0.05 M Bu4NBr solution at 80  $\rm{^oC}$  for several minutes, only the treated side showed the adhesive property. In control experiments, films of **3** did not become adhesive after they were immersed in aqueous 8 M NaOH solution alone or in **0.05** M Bu<sub>4</sub>NBr solution at 80 °C for several days. Thus, Bu<sub>4</sub>NBr appears to increase the effectiveness of the reaction, probably by enhancing the nucleophilicity of the hydroxide anion. However, the surfactant properties of  $Bu$ <sub>A</sub>NBr may also contribute to the catalysis.

Finally, it should be noted that the surface reaction of **3** could also be carried out in 8 M NaOH/0.05 M Bu4NBr solutions at room temperature but that the rate of the surface reaction was noticeably slower than at 80 °C. Adhesive surfaces of **3** similar to those detected in the 80 "C experiments were apparent after **9** days at room temperature. Polymer **3** undergoes a softening transition at about 80 "C, the temperature employed for the surface reactions. However, the reaction at room temperature shows that **3** need not be heated to the softening transition to induce the surface reaction. But softening of the polymer probably increases the rate of the surface reaction. Although the effect of temperature on the rate of surface hydrolysis was not quantified, it seems clear that temperature variations can be used to control the rate of the reaction. Similarly, it appears that the rate of surface modification can be controlled by variations in the concentration of NaOH and/or Bu4NBr. Reactions to generate adhesive surfaces were also carried out with KOH/Bu4NBr or Bu4NOH solutions.

Surface Analysis of the Modified **Films.** On the basis of the solution reactions of the model compound **5**  (Scheme II), it was suspected that  $Bu<sub>4</sub>NBr$  facilitates a nucleophilic replacement of trifluoroethoxy side groups by  $-O<sup>-</sup>NBu<sub>4</sub><sup>+</sup>$  groups at the film surface (Scheme III). If the reaction illustrated in Scheme I11 does occur, then the modified films should undergo exchange reactions with cationic dyes but not with anionic dyes. This was found to be the case.

Surface-treated films of **3** that were immersed in aqueous solutions of the cationic dye methylene blue instantly adopted the dye's blue color and retained the color after extended immersion in pure water to remove loosely adsorbed dye. However, immersion of the treated films of **3** in aqueous solutions of the anionic dye Biebrich scarlet resulted in no permanent color change. Control experiments showed that both untreated films of **3** and films of **3** that had been treated with 8 M NaOH alone or in 0.05 M Bu4NBr solutions at *80* "C did not become colored when immersed in the cationic or anionic dyes.

The staining by methylene blue allowed the approximate depth of the surface reaction to be measured by optical microscopic examination of a cross section of the dyed film.16 The depth of reaction was estimated by means of a calibrated eye piece. The data are illustrated in Figure



**Figure 1.** Depth of surface hydrolysis of  $[NP(OCH_2CF_3)_2]_n$  as a function of reaction time.



Figure **2.** Attenuated total internal reflectance infrared spectra of the surface of  $[NP(OCH_2CF_3)_2]_n$  (A) before surface hydrolysis and (B) after treatment for 1 min with an aqueous 8 M NaOH/0.05 M Bu4NBr solution at 80 "C. The assignments for absorption **peaks** of interest are (1) OH or H20, **(2)** aliphatic C-H stretching mode from  $CH_2$  units in  $OCH_2CF_3$  and butyl units, (3)  $Bu<sub>4</sub>N<sup>+</sup> units$ , (4) the phosphazene skeleton, and (5)  $CF<sub>3</sub>$  groups.

1 and show a nonlinear increase in the depth of modification **as** a function of reaction time. The reduction in penetration rate can be attributed to a decrease in the concentration of the active species,  $Bu<sub>4</sub>NOH$ , at increasing distances from the surface due to diffusion effects. In addition, replacement of trifluoroethoxy side groups by tetra-n-butylammonium hydroxyl side groups probably becomes more difficult as the reaction progresses (see Schemes II and III and later comments).

The ATR-IR spectra of films of **3** before reaction and after **1** min of reaction in 8 M NaOH/0.05 M Bu4NBr solutions at 80 "C are shown in Figure **2.** The ATR-IR spectra are sensitive to the composition of *approximately*  the top  $1 \mu m$  of the polymer surface.<sup>17</sup> On the basis of the depth profiling data in Figure **2,** the surface reaction should be detectable by this method. Spectrum A of the pristine film showed aliphatic C-H stretches at **2950** cm-l for the methylene unit of the trifluoroethoxy groups. Enhanced absorptions at 2850-3000, 1647, and 1300-1600

**<sup>(16)</sup> Gibson, H. W.; Bailey, F. C.** *Macromolecules* **1980,** *13,* **34.** 

**<sup>(17)</sup> Harrick, N.** J. *Internal Reflection Spectroscopy;* **Interscience: New York, 1967.** 



**Binding Energy (eV)** 

**Figure 3. X-ray photoelectron spectrum of the surface of [NP-**   $(OCH_2CF_3)_2$ , before (lower spectrum) and after surface hydrolysis, **showing the C 1s binding energies.** 

cm-' were detected in the ATR-IR spectrum of the treated film (spectrum B), and on the basis of the transmission IR spectrum of  $Bu_4$ NBr, these were assigned to tetra-nbutylammonium hydroxyl groups present on the film surface. It is noteworthy that large amounts of surface hydroxyl groups  $(3500-3200 \text{ cm}^{-1})$  or P-OH groups  $(2700$ cm-') were not detected in the ATR-IR spectrum B. The broad, low-intensity absorption at  $3200-3400$  cm<sup>-1</sup> in spectrum B was tentatively assigned to adsorbed water associated with the hygroscopic tetra-n-butylammonium sites on the film surface. Little change in the ATR-IR spectra was detected with longer reaction times. No significant changes in the ATR-IR spectra below  $1300 \text{ cm}^{-1}$ were detected following the surface reaction.

X-ray photoelectron spectroscopic (XPS) analysis was also carried out. In contrast to the ATR-IR measurements, the XPS measurements are sensitive to the composition of the outer 50-100 **A** of the film surface.ls XPS spectra of untreated 3 and the fluoroalkoxyphosphazene elastomer **4** have been reported in the literature.<sup>2,19</sup> In the present work, the untreated polymer yielded one 0 1s peak (535 eV), one N 1s peak (400 eV), one P 2p peak (136 eV), one F 1s peak (689 eV), and two C 1s peaks (294,289 eV), and this is consistent with the bulk structure of **3.** It was necessary to wash films of **3** in hexane to remove traces **of** silicon-containing contaminants (probably organosiloxanes from stopcock grease) that had migrated to the surface.

After the surface treatment, films of **3** were immersed in distilled-deionized water for several minutes to remove loosely adsorbed species, and the films were dried under



Figure 4. Hydrophobicity of the surface of  $[NP(OCH_2CF_3)_2]_{n_2}$ **as measured by water droplet contact angle, as a function of reaction time when** films **of the polymer were** treated **with aqueous**  8 M NaOH/0.05 M Bu<sub>4</sub>NBr at 80 °C.

vacuum. The surface elemental analysis of the modified films after 1 min of treatment with the 8 M NaOH/0.05 M Bu4NBr solution at 80 "C indicated an 8% increase in carbon and a **7%** decrease in fluorine concentrations at the surface. This is consistent with the reaction shown in Scheme 111. The XPS analysis suggested an approximate surface formula of  $[NP(OCH_2CF_3)_{1.80}(OBu_4N)_{0.20}]_n$ .

In addition to the change in elemental composition at the surface, a new C 1s peak at 285 eV and a new N 1s **peak**  at 405 eV were detected after the reaction. These new peaks were assigned to the tetra-n-butylammonium cation. The C 1s spectra before and after the surface reaction are shown in Figure 3. No new 0 ls, P 2p, or F 1s peaks were detected in the *XPS* spectra after the reaction. In control experiments, no major changes in the XPS spectra were detected when films of **3** were immersed in 8 M NaOH alone or in 0.05 M Bu<sub>4</sub>NBr solutions at 80  $\rm{^{\circ}C}$  for 1 min.

It is important to note that the XPS spectra of the films after the reaction showed that significant amounts of sodium (from NaOH) or bromine (from Bu<sub>4</sub>NBr) were not present in the **film** surface. In addition, elemental analysis for sodium and bromine of bulk films of **3** that had been hydrolyzed in 8 M NaOH/0.05 M Bu<sub>4</sub>NBr at 80 °C for 24 h and washed with water **also** suggested that sodium and bromine were not incorporated into the modified films.

Changes in the surface hydrophilicity of **3 as** a function of immersion time in the 8 **M** NaOH/0.05 M Bu4NBr reagent at 80 °C were measured by contact-angle analysis (using drops of unbuffered, distilled-deionized water, pH  $= 5.6$ .<sup>20</sup> Contact-angle measurements are sensitive to the composition of the outer **5-10** *8,* of a surface. Untreated films of **3** are hydrophobic, with a water contact angle of 108<sup>o</sup>. Replacement of trifluoroethoxy side groups by ionic tetra-n-butylammonium hydroxyl side groups (Scheme 111) would be expected to increase the hydrophilicity. **This was**  found to be the case (Figure **4).** 

The variation of contact angle **as** a function of immersion time in the 8 M NaOH/0.05 M Bu<sub>4</sub>NBr solution at 80 °C is shown in Figure 4. Although the surface became more hydrophilic following the initial treatment, exposure to the reagent solution for more than 1 min did not bring about

<sup>(18)</sup> Gardella, J. A. Appl. Surf. Sci. 1988, 31, 72.<br>(19) Hiraoka, H.; Lee, W. Y.; Welsh, L. W.; Allen, R. W. Macromole-<br>cules 1979, 12, 753.

**<sup>(20)</sup> Yasuda, H.; Sharma, A. K.** *J. Polym. Sci., Polym. Phys. Ed.* **1981, 19, 1285.** 

a further increase in the surface hydrophilicity. It is perhaps surprising that the minimum contact angle was 90°, and this indicated that appreciable hydrophobic character remained. This residual surface hydrophobicity can be explained by the presence of  $n$ -butyl and/or trifluoromethyl groups in the top 5-10 **A** of the surface or by the fact that the extent of the surface reaction is quite low. Although water does not progressively wet the surface of untreated 3 over a period of time, the surface of modified 3 was progressively wetted over a period of 15 min. This suggested that water interacts with and perhaps reacts with the modified surface. Although surface roughness can affect contact angle measurements, SEM analyses of the modified surfaces showed that the surface texture did not change.

The contact angles of water on the modified surfaces were found to be insensitive to the acidity or basicity of the water drop (pH of **3** or 10, buffered). This suggests that, although an anionic hydroxyl surface is formed by the reaction, instantaneous acid-base reactions do not occur at the solid-liquid interface. This contrasts with the behavior of carboxylic acid groups generated by surface oxidation of polyethylene<sup>21</sup> or sulfonic acid groups in surface-sulfonated polystyrene.

**Attempted Reconstruction** of **the Modified Surface.**  It is known that polymer surfaces may reorganize in different environments or at different temperatures.<sup>2,22</sup> Typically, if a polymer is exposed to a polar environment, polar functional groups may migrate to the surface. In a nonpolar environment, nonpolar functional groups in the polymer will migrate to the surface. These reorganizations may or may not be reversible, and the thermodynamic and kinetic factors that govern the reversibility of surface reconstruction are only poorly understood. A characteristic feature of phosphazene polymers is the facile torsional mobility of the polymer backbone. For example, the  $T<sub>g</sub>$ of 3 is **-66** "C. Thus, a room temperature, 3 is well above its *Tg,* and the polymer backbone can undergo torsional motions that should facilitate a surface reconstruction. Reconstruction should also be facilitated by heating the polymer to the phase transitions at 80 "C.

The reacted polymer surface shown in Scheme III is potentially amphiphilic, since both the ionic tetra-n-butylammonium hydroxyl side groups and the hydrophobic trifluoroethoxy side groups are present. In the experiments described in the previous sections, the reacted films of 3 were removed from the aqueous NaOH/Bu<sub>4</sub>NBr reagent, rinsed with distilled-deionized water for several minutes, and dried under vacuum. No change in adhesion (as defined previously) was apparent when modified films of 3 were exposed to the air or water over extended periods of time.

It was anticipated that the surface **free** energy of the film would perhaps be minimized if low surface energy trifluoroethoxy groups were induced to replace tetra-n-butylammonium groups at the **film** surface by heat treatment or by placement in a nonpolar environment (see Experimental Section). However, surface-treated films of 3 retained their adhesive character after being heated to 80 **OC** under vacuum. Similarly, when the reacted films of 3 were immersed in boiling hexane, the adhesive surface properties remained. In these experiments, surface reconstruction to form a hydrophobic surface composed principally of trifluoroethoxy side groups could not be



Figure **5. lH** NMR spectrum of **an** acetone solution made from a film of  $[NP(OCH_2CF_3)_2]_n$  treated for 24 h with aqueous 8 M  $NaOH/0.5$  M Bu<sub>4</sub>NBr at 80 °C.

detected by ATR-IR analysis, by contact-angle analysis, or by exchange reactions with cationic dyes.

Surface-treated films of 3 immersed in ethanol lost their capacity to adhere to metals polymers or ceramics. However, it was clear that ethanol slowly dissolved the surface material from the bulk of the film and that the loss of adhesive character was not due to reconstruction. Despite these changes, it should be noted that ATR-IR analysis, contact-angle analysis, and exchange reactions with cationic dyes indicated that some tetra-n-butylammonium hydroxyl groups were still present on the film surface after immersion in ethanol.

Several reasons for the apparent lack of reconstruction of the modified surfaces can be given. The process may not be thermodynamically favored despite the ionic character of the surface, due to the hydrophobic character of butyl groups or the relatively **small** differences in surface energies. The ionic character of the hydroxyl side groups may become sufficiently important to drive a reorganization only when the ions are dissociated. Also, even if the reconstruction process is favored thermodynamically, the process may be blocked kinetically. For example, the stronger mechanical integrity of the bulk of the polymer relative to the surface could, in principle, slow a reconstruction process.

**NMR and GPC Analysis of the Surface-Modified Films.** It seemed likely that a prolonged surface reaction might be sufficiently pervasive to allow classical solution methods of analysis, such **as** NMR, to be applied to the system. Thus, the surface reaction of 3 in 8 M NaOH/0.05 M Bu4NBr solution at 80 "C was allowed to proceed for **24** h, and the modified films were then immersed in water for several hours and were dried under vacuum. The **films**  were gellike and retained little mechanical strength. This prevented an accurate measurement of the depth of reaction by dye exchange. However, the films were still completely soluble in THF or acetone. The 'H NMR spectrum of the modified film dissolved in acetone- $d_6$  is shown in Figure 5.

The 'H NMR spectrum of unreacted 3 consists of a methylene resonance for the trifluoroethoxy groups at  $\delta$ **4.6.** The 'H NMR spectrum for the hydrolyzed film (Figure 5) consisted of two seta of trifluoroethoxy resonances  $(\delta 4.6, 4.2)$ , four resonances for the *n*-butyl protons (6 1.0,1.5, 1.9, and **3.4),** and a resonance for residual water (6 3.3). **This** water could not be removed completely under vacuum, and its presence was attributed to the hygroscopic nature of tetra-n-butylammonium salt units. As mentioned previously, water was also detected from the

**<sup>(21)</sup> Holmes-Farley, S. R.; Bain, C. D.; Whitesides, G. M.** *Langmuir*  **1988,** *4,* **921.** 

**<sup>(22)</sup> Gagnon, D. R.; McCarthy, T.** J. *J. Appl. Polym. Sci.* **1984,** *29,*  **4335.** 



**Figure 6.** Plot of molecular weight versus time for the polymer obtained by the heterophase treatment of a film of [NP-  $(OCH_2CF_3)_2]_n$  with an aqueous solution of 8 M NaOH/0.05 M Bu4NBr at 80 **OC.** The molecular weight values were obtained by gel permeation chromatography with the system calibrated by polystyrene standards.

## ATR-IR spectra **of** the modified film surfaces.

The 6 **4.6** resonance was assigned to trifluoroethoxy side groups geminal to another trifluoroethoxy structure, while the  $\delta$  4.2 peak was assigned to trifluoroethoxy side groups that were adjacent to geminal tetra-n-butylammonium hydroxyl side groups.14 The degree of replacement of trifluoroethoxy side groups by tetra-n-butylammonium hydroxyl side groups could be estimated by integration of the trifluoroethoxy and the n-butyl resonances. Approximately **34%** of the side groups were tetra-n-butylammonium hydroxyl units after **24** h **of** hydrolysis.

The 13C NMR spectrum of unreacted **3** consisted of a quartet centered at 6 **63** for the methylene carbon of the trifluoroethoxy group and a quartet centered at 6 **120** from the  $CF_3$  carbon of the trifluoroethoxy group. The <sup>13</sup>C NMR spectrum of the modified polymer consisted of the four n-butyl carbon resonances (6 **12,19,23,** and **57)** and the two trifluoroethoxy quartets.

The 31P NMR spectrum of **3** in acetone before reaction consists of a singlet at  $\delta$  -4. However, the <sup>31</sup>P NMR spectrum of the modified polymer consisted **of** a singlet at  $\delta$  -4 plus two minor broad resonances at  $\delta$  -1 and -6. The sharp singlet was attributed to unreacted  $P(OCH<sub>2</sub>Cl<sub>-</sub>)$  $\mathbf{F}_3$ )<sub>2</sub> units, and the broad resonances were assigned to hydrolyzed units. The trifluoroethoxy fluorine atoms ( $\delta$ 39.3) were detected in the <sup>19</sup>F NMR spectra of both the unmodified and modified films. Thus, the NMR spectra were consistent with the belief that tetra-n-butylammonium hydroxyl side groups had replaced some of the trifluoroethoxy side groups at the surface (Scheme 111).

The solubility of the grossly modified films **also** allowed analysis by gel permeation chromatography (GPC). In Figure **6,** the GPC average molecular weight **of** the modified polymer is shown **as** a function of immersion time in 8 M NaOH/0.05 M Bu<sub>4</sub>NBr solution at 80 °C. A slight but statistically significant decrease in molecular weight was detected with increasing reaction time.

Although some change in hydrodynamic volume may occur with the hydrolysis reaction and this may affect the GPC measurements, changes were also found in the molecular weight distribution of the polymer. A bimodal molecular weight distribution was present before and **after**  the reaction. However, after the reaction, the proportion

Scheme IV



of the higher molecular weight fraction was reduced relative to that of the lower molecular weight component. Replacement of the trifluoroethoxy side groups **(99** g/mol) by tetra-n-butylammonium hydroxyl side groups **(258**   $g/mol$ ) would increase the  $M_w$  slightly. In control experiments, no change in molecular weight was detected by GPC analysis when **films** of **3** were immersed in 8 M NaOH only or in 0.05 M Bu4NBr solutions at 80 "C for **24** h.

**Hydrolysis of Solutions of 3 by NaOH/Bu4NBr in THF.** The hydrolysis reaction of solutions of **3** by NaOH in the presence of Bu4NBr in refluxing THF **(66** "C) was also investigated in order to provide further information about possible reactions that may occur during the surface reaction. Polymer **3** was dissolved in THF, NaOH and Bu4NBr were added, and the reaction mixture was heated to reflux and monitored by 31P NMR spectroscopy.

The 31P NMR spectrum of **3** in THF solution consists of a singlet at  $\delta$  -8. A new resonance appeared at  $\delta$  -6 as hydrolysis progressed. This resonance was also detected in the surface modified films and was assigned to  $P(OCH_2CF_3)(O-Bu_4N^+)$  phosphorus atoms. Consistent with the model-compound studies (Scheme 11), the spectra showed that the hydrolysis reaction proceeded by a nongeminal mechanism. After **90** min **of** hydrolysis, the **31P**  NMR spectrum still consisted only of the  $\delta$  -6 and -8 resonances. Hydrolysis of **3** by NaOH/Bu4NBr for **9** h in refluxing THF generated broad peaks in the 31P NMR spectrum over a range  $\delta$  4 to  $-\delta$ .

After **90** min of hydrolysis, analysis of the 31P NMR spectrum indicated that approximately **20%** of the trifluoroethoxy side groups had been replaced by tetra-nbutylammonium hydroxyl side groups (Scheme **IV).** This polymer is designated as structure **6.** Polymer **6,** when isolated from solution, was gellike, adhesive, and stable to the atmosphere over extended periods of time. It did not cross-link, despite the presence of the hydroxyl side groups.

The 'H **NMR** spectrum of **6** matched that of the surface modified films. According to the 'H NMR analysis, water was present in **6** despite drying, which indicated that **6** was hygroscopic (as were the surfaces of the modified films). In addition, the <sup>13</sup>C and <sup>19</sup>F NMR spectra of 6 were identical with the spectra of the surface-reacted films. The transmission IR spectrum of **6** was **also** very similar to that of the surface-modified material and showed the characteristic absorptions of the tetra-n-butylammonium hydroxyl side groups.

GPC analysis of **3** before the solution hydrolysis reaction indicated an  $M_{\rm w}$  of  $3.9 \times 10^5$ . After hydrolysis for 90 min, a GPC analysis of **6** indicated a slightly lower *M,* of **2.4**   $\times$  10<sup>5</sup>. Although the change was small, the result suggests that a limited amount of chain cleavage may occur in the solution hydrolysis **as** well **as** during the surface hydrolysis. From these data, it appears that the reactions at the surface of **3** resemble closely those **of 3** in solution.

**Surface Treatment of Fluoroalkoxyphosphazene Elastomers and Other Phosphazene Polymers.** The surface reactions of a commercial fluoroalkoxyphosphazene elastomer **(4)** was examined briefly. This elastomer is composed of approximately **65** % trifluoroethoxy side groups and **35%** mixed, longer chain fluoroalkoxy side groups. The commercial product is also cross-linked and contains additive materials. Hence, differences in the rates

of surface reaction for **3** and **4** might be expected. When samples of 4 were immersed in 8 M NaOH/0.05 M Bu. NBr solution at 80 °C for 6 h, the surfaces became adhesive, although the changes were much slower than for polymer **3.** 

Films of the phosphazene polymers poly(diphenoxyphosphazene) **(7),** poly **[bis(p-bromophenoxy)phosphazene]** 



immersed in 8 M NaOH/0.05 M Bu<sub>4</sub>NBr solutions at 80 **OC** for six hours. No surface reactions were detected visually **or** by ATR-IR analysis, and the surfaces did not adhere to metals, ceramics, or other materials. In addition, polymers **7-9** did not react with NaOH/Bu4NBr in *solution* in refluxing THF after **3** h. The phenoxy analogue of 3,  $[NP(OC_6H_5)_2]_n$ , has been reported to be resistant to hydrolysis by NaOH under conditions that will induce the hydrolysis of **5.23** Thus, the trifluoroethoxy side group appears to provide a special opportunity for surface reactions with polyphosphazenes. This is a consequence of its high reactivity as a leaving group.

**Mechanism of Adhesion.** In the past 50 years, many theories have been proposed to explain adhesion, and numerous interrelated physical, chemical, mechanical, and rheological factors have been identified that appear to influence the adhesion of materials. $24-26$  Much disagreement still exists about the relative importance of each factor **for** specific systems, since it is difficult to analyze an adhesive joint nondestructively or to isolate the influence of one factor from others. A survey of the literature indicates that the adhesive properties of phosphazene polymers is an area that has not yet been explored extensively.

Several processes occur at the polymer surface when films of **3** are immersed in hot, aqueous NaOH/Bu4NBr solutions. These include the generation of ionic, hygroscopic tetra-n-butylammonium hydroxyl side groups, formation **of** a gellike surface, an increase in surface hydrophilicity, and limited chain cleavage. All of these changes can, in principle, contribute to adhesion of the reacted films to the surfaces of common substrates such as metals, glasses, polymers, ceramics, or living tissue.

The gellike character of the surface could allow the polymer chains to penetrate into the microcrevices of the substrate surface. Ionic groups at the surface increase the hydrophilicity and allow the polymer to wet the surface

of a polar substrate. In addition, the ionic hydroxyl side groups can hydrogen-bond to hydroxyl groups on the substrate surface and can generate ion-dipole/ion-ion forces at the interface. Thus, at a qualitative level, the adhesive properties of the surface-hydrolyzed phosphazene polymer *can* be rationalized in terms of molecular structure and current views concerning adhesion.

Experiments indicated that the cation associated with the hydroxyl side group at the surface of the reacted films plays a role in determining the adhesive behavior. For example, when the surface modified films were soaked in aqueous solutions of sodium, potassium, calcium, magnesium, copper(II), or iron(II1) chloride for **24** h at **50 "C,** the films lost their adhesive character. Furthermore, films soaked in solutions of the divalent cations were clearly more brittle than those exposed to solutions of the monovalent cations. This presumably reflects an ionic cross-linking by the divalent or trivalent metal cation at the surface. ATR-IR experiment analysis after this treatment suggested that the tetra-n-butylammonium groups were no longer present. Adhesive films could also be generated by the surface modification of **3** in aqueous potassium hydroxide solutions with 18-crown-6 ether present as phase transfer agent.

#### **Conclusions**

The linkage of tetra-n-butylammonium hydroxyl groups to the surface of other polymers may generate adhesive character for these species **as** well. Polymers that possess quaternary ammonium salt side groups have shown enhanced antibacterial activity, $^{27}$  and anionic sites at polymer surfaces typically improve the blood compatibility of a polymer.2 Further experiments to modify the surfaces of **3** with other nucleophiles and to test effects of surface reactions on the biological properties of **3** are in progress.%

In general, a need exists to develop the surface chemistry of phosphazene polymers, and it is significant that the reactions that take place at the surface of **3** are very similar *to* those that occur in solution. **Thus,** the solution reactions of phosphazene polymers appear to "model" the surface reactions of the same material. The generation of hydroxyl side groups also offers opportunities for further functionalization of the polymer surfaces.

#### **Experimental Section**

**Materials. Compound 1 was provided by Ethyl Corp. (Baton Rouge, LA) and was recrystallized from hexane before use. Polymer 2 was obtained by the thermal ring-opening polymerization of 1 at 250 "C. Tetra-n-butylammonium bromide (Aldrich) and 18-crown-6 ether (Aldrich) were used as received. Tetra-nbutylammonium hydroxide (Aldrich) was obtained as a 40% aqueous solution. Biebrich scarlet (Aldrich) and methylene blue (Aldrich) were used as received. Distilled water was passed through a Millipore ion-exchange resin before use. Commercial**  fluoroalkoxyphosphazene O-rings 4 were provided by Ethyl Corp. THF **and toluene were distilled** from **sodium benzophenone under a nitrogen atmosphere. The syntheses of 3,5 7,b and 829 are described elsewhere.** 

**Instrumentation and Methods. NMR spectra were obtained with the use of either a JEOL FX9oQ or a Bruker WP-360 NMR spectrometer, with slP chemical shifts referenced to aqueous 85%**  phosphoric acid and <sup>1</sup>H spectra to acetone- $d_6$ . IR spectra were **obtained with use of a Perkin-Elmer 1710 FT-IR spectrometer. GPC data were obtained by means of a Hewlett-Packard HP1090 liquid chromatograph equipped with Polymer Lab PLGel columns** 

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### *Poly[bis(trifluoroethoxy)phosphazene] Films*

**(108, 108, 109** A) and a refractive index detector. The mobile phase was a **0.1%** solution of tetra-n-butylammonium bromide in THF. The GPC columns were calibrated with narrow molecular weight polystyrene standards.

ATR-IR spectra were obtained with the Perkin-Elmer **1710**  FT-IR spectrometer equipped with an ATR attachment (Harrick Scientific Corp., Ossining, NY). A germanium (Ge) internal reflection element (45<sup>°</sup>) was positioned at a 45<sup>°</sup> angle with respect to the incident IR beam. Both faces of the Ge crystal were covered with polymer film. The intensity of the IR beam at the detector was maximized by adjusting the positions of the Ge crystal and the reflecting mirrors. One hundred scans were averaged. A resolution of **2** cm-' was employed. The Ge crystal was cleaned with carbon tetrachloride before and after use.

Contact-angle measurements were obtained with a Rame-Hart contact angle goniometer equipped with a telescope, an enviflat-tipped needle. The polymer films were mounted on micro-<br>scope slides by means of double-sided adhesive tape. Polymer films were placed in the environmental chamber, the atmosphere of which was saturated with water vapor. A  $1-\mu L$  drop of un-<br>buffered, distilled-deionized water was suspended from the syringe needle above the film surface, and the needle was lowered carefully until the water drop was placed onto the polymer surface. The needle was raised carefully, and the contact angle measurement was made on both sides of the drop (within **30** s). The contact angle values reported are averages of five measurements made on different areas of the film surface. Scatter in the data was *5O* or less.

X-ray photoelectron spectra (XPS) were recorded at the Perkin-Elmer Physical Electronics Laboratory (Edison, NJ) on a Perkin-Elmer-Physical Electronics **5000** LS spectrometer. A monochromatic X-ray source using an Al anode (Al  $K\alpha = 1486.6$  eV) operated at 600 W (15kJ, 40 mA) was employed as the primary excitation source. Charge compensation was accomplished by flooding the sample surface with low-energy **(<2** eV) electrons. Spectra were recorded with a 65° takeoff angle. Atomic composition data were determined by using the instrument's computer and programmed sensitivity factors. Scanning electron microscopic analysis was performed with the use of an **IS1 SX-4OA**  instrument.

**Adhesion Testing.** A film of 3 was immersed in 8 M NaOH/0.05 M Bu4NBr solution at *80* "C for **5** min. The film was rinsed with distilled-deionized water for **5** min and dried under vacuum for **1** h. The adhesive film was pressed with Teflon forceps onto degreased aluminum foil. The remaining side of the adhesive film was attached to a glass plate via double-sided adhesive tape.

Strips of aluminum foil **3.2** cm wide were cut with a razor blade, and a tab was carefully initiated. Tabs were puled with **an** htron **4201.** The peel angle was **90°,** and data were collected when the films were between 85 and 95°. Tabs were pulled at a crosshead speed of **10.00** mm/min. Four runs were averaged, and values of **14.1, 23.2, 19.1,** and **22.6** mN/mm were obtained.

**Surface Reaction Experiments.** The aqueous NaOH/<br>Bu<sub>a</sub>NBr solutions (typically 30 mL) were prepared in a clean glass jars, and the solutions were heated to 80 °C with an oil bath. Polymer films were handled at the edges with Teflon-coated forceps. It should be noted that at **an** 8 M concentration of NaOH, much of the Bu4NBr is insoluble in the aqueous solution, and the insoluble material floats to the surface of the solution. The films were immersed below the surface of the NaOH/Bu.NBr solutions for the required period of time. They were then quickly removed, gently rinsed with and immersed in distilled-deionized water for several minutes, and dried under vacuum for at least an hour.

**Depth Profiling Data.** Modified films of 3 were immersed in a  $2 \times 10^{-2}$  M aqueous solution of methylene blue for 48 h at room temperature. The dyed films were then immersed in distilled-deionized water for **48** h until no additional free dye was removed from the films. Cross sections of the films were examined with a light microscope **(1500X)** equipped with a calibrated eye

piece. **Reaction of 3 with NaOH/Bu4NBr in THF.** Polymer **3 (1.0**  g, 0.17 mmol P-Cl) was dissolved in 15 mL of THF, and Bu<sub>4</sub>NBr **(1.0** g, **3.1** mmol) and NaOH **(1.0** g, **25** mmol) were added. The heterogeneous mixture was heated to reflux with an oil bath, and aliquots of the solution were removed for 31P NMR analysis. The isolation of **6** consisted of decanting the THF solution from excess NaOH, concentration of the solution by rotary evaporation, and precipitation of the polymer into distilled-deionized water (four times) to remove salts. Polymer **6** was dried under vacuum.

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Registry **No. 8** (SRU), **52734-85-7; 9** (SRU), **133294-79-8;**  [NP(OCH,CF,),], (SRU), **28212-50-2;** BuN+Br-, **1643-19-2;**  [Np(Ph),], (SRU), **28212-48-8;** methylene blue, **61-73-4;** Biebrich scarlet, **4196-99-0.**